## Electronic Properties of the NREL Low Filament Temperature HWCVD Amorphous Silicon Germanium Alloys

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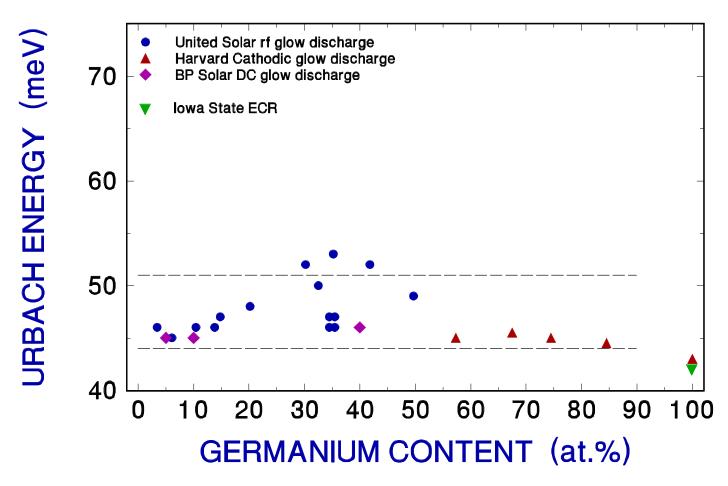
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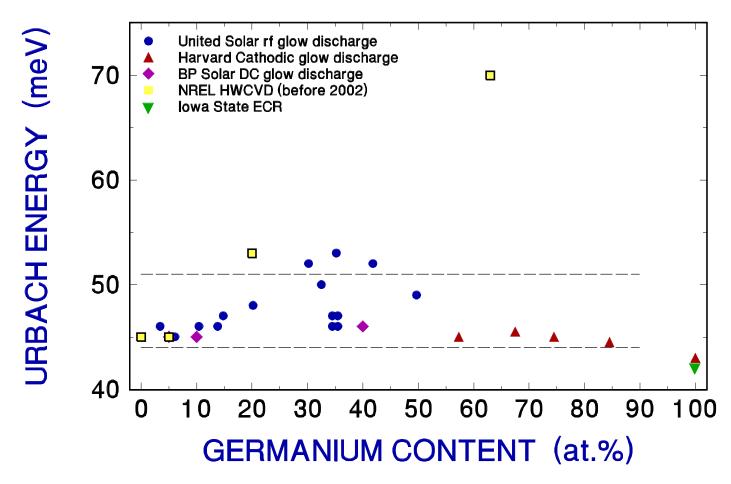


## Urbach Energies for the Best a-Si,Ge:H





## The Best PECVD vs. HWCVD a-Si,Ge:H

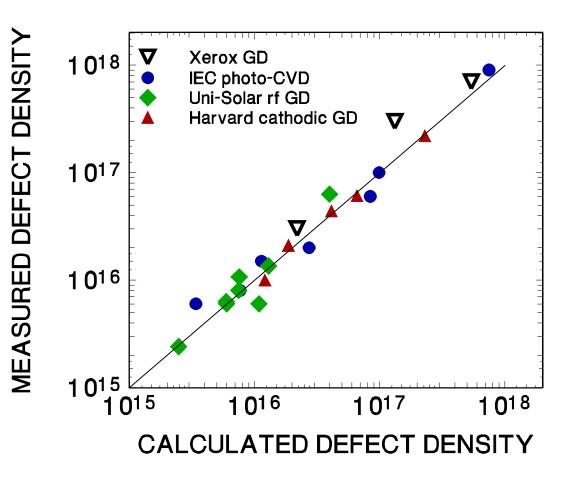




# Urbach Energy plus Gap Energy Determines Defect Density

$$N_D = N_0 E_U \exp[-(E_D - E^*)/E_U]^{-1}$$

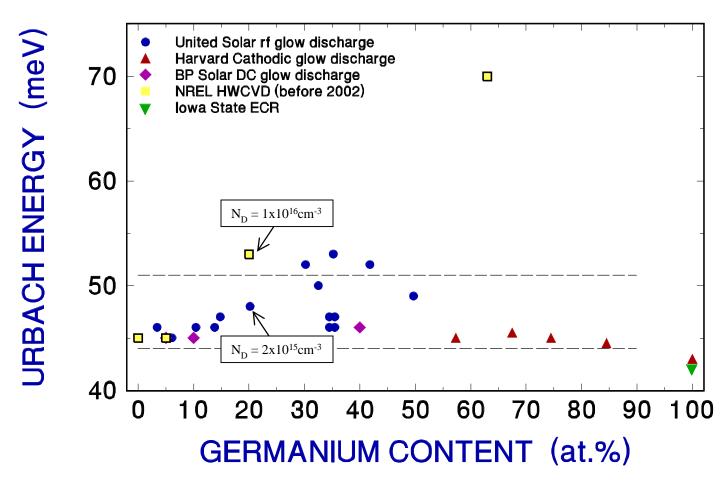
- •• Single choice of N<sub>0</sub> and E\* used for all sets of samples
- The germanium fraction of these alloys varies between 0.2 and 1.0.





<sup>&</sup>lt;sup>1</sup> M. Stutzmann, Philos. Mag. B**60**, 531 (1989)

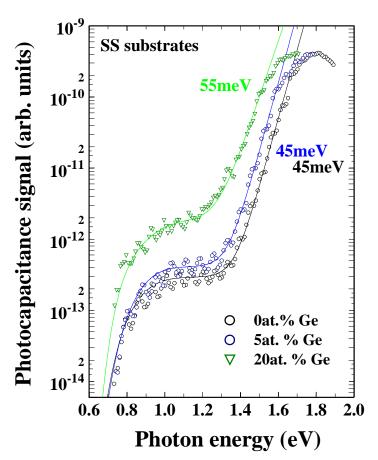
## The Best PECVD vs. HWCVD a-Si,Ge:H





## Spectra for NREL HWCVD a-Si,Ge:H

#### 2000°C Tungsten Filament



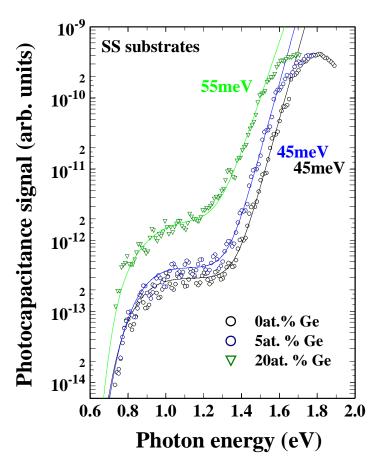
Best HWCVD
Samples prior to 2004

Does this mean that Ion Bombardment is crucial for high quality a-Si,Ge:H?

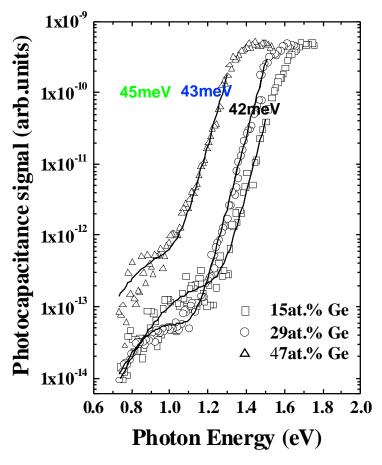


## Spectra for NREL HWCVD a-Si,Ge:H

#### 2000°C Tungsten Filament

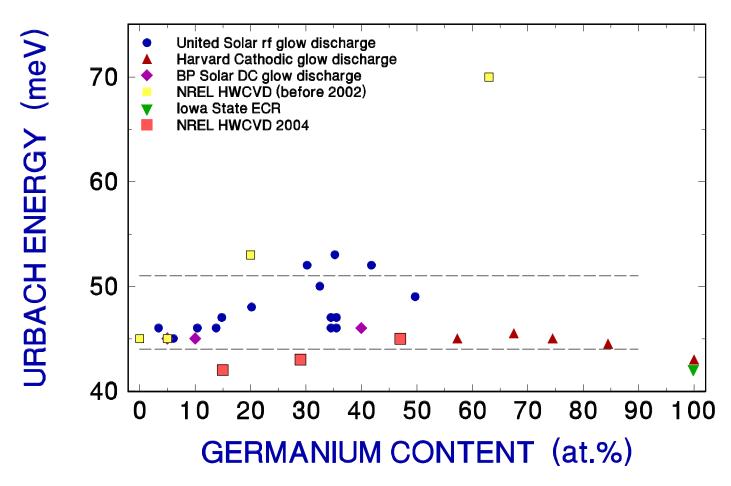


#### 1800°C Tantalum Filament





#### The Best PECVD vs. HWCVD a-Si,Ge:H

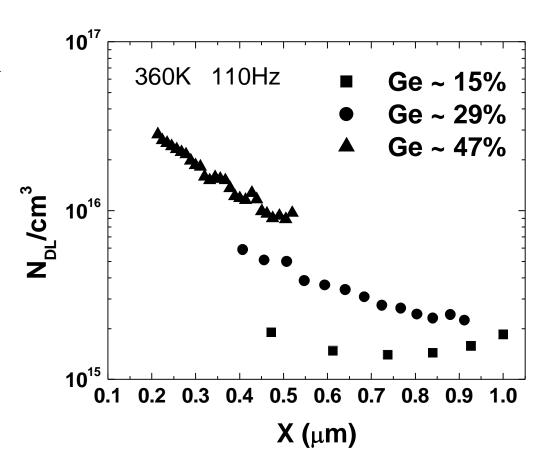




## Mid-gap Defect Density Profiles via DLCP

Note that  $N_{\rm D}$  increases in the direction of film growth

This reflects the fact that the substrate temperature increased as much as 90°C during growth





## Properties of New HWCVD a-Si,Ge:H Alloys

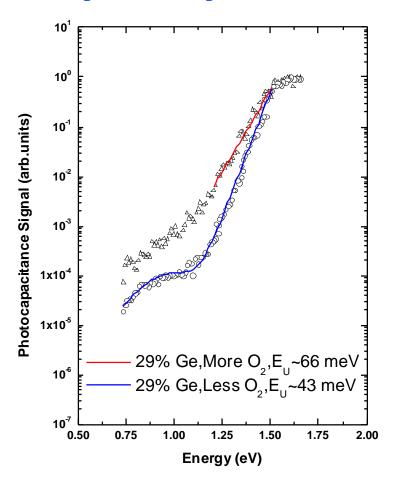
## New NREL HWCVD a-Si,Ge:H alloys exhibit electronic properties as good as any alloys ever characterized

Sample	Ge Fraction (at.%)	Growth Rate (Å/s)	E <sub>04</sub> (eV)	E <sub>Tauc</sub> (eV)	Eu (meV)	Defect Density (cm-3)
L1305	15	1.39	1.79	1.65	$42\pm2$	2 × 10 <sup>15</sup>
L1306	29	1.78	1.66	1.50	$43\pm3$	$4\pm2\times10^{15}$
L1307	47	2.00	1.47	1.32	$45\pm 2$	2±1 × 10 <sup>16</sup>

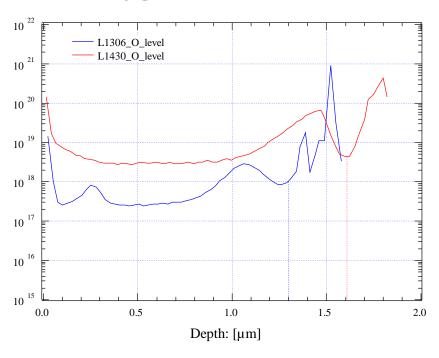
- •Urbach energies from Photocapacitance Spectra
- •Midgap Defect densities from DLCP measurements
- Note relatively high growth rates

## Recent Samples Exhibit Poorer Properties

#### Photocapacitance Spectra: 29at.% Ge



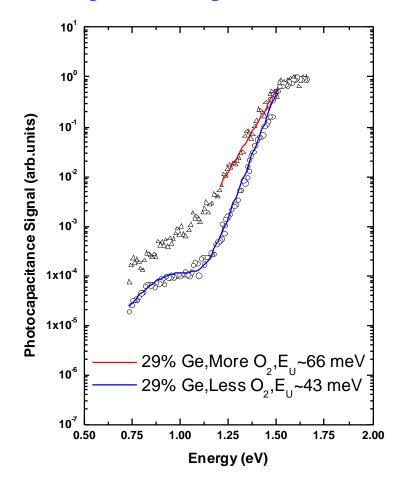
#### Oxygen SIMS Profiles

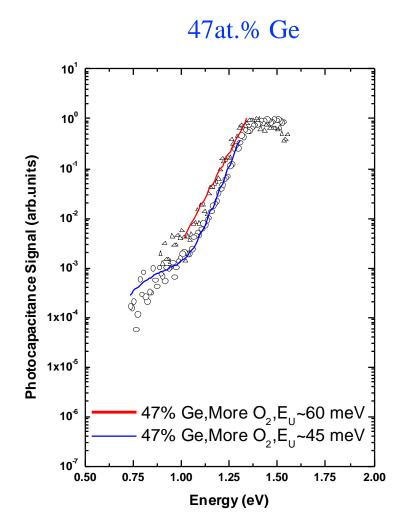




## Recent Samples Exhibit Poorer Properties

Photocapacitance Spectra: 29at.% Ge

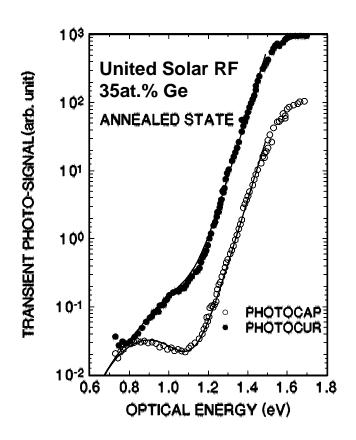


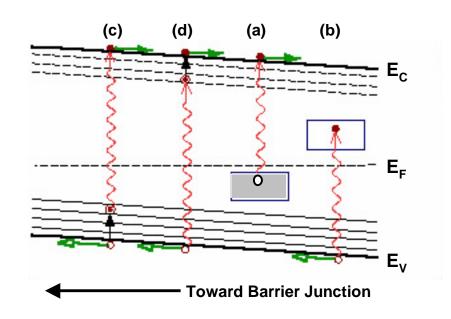




## Measuring Fraction of Collected Holes

#### Obtained from Ratio of TPI to TPC signals in Bandtail Region



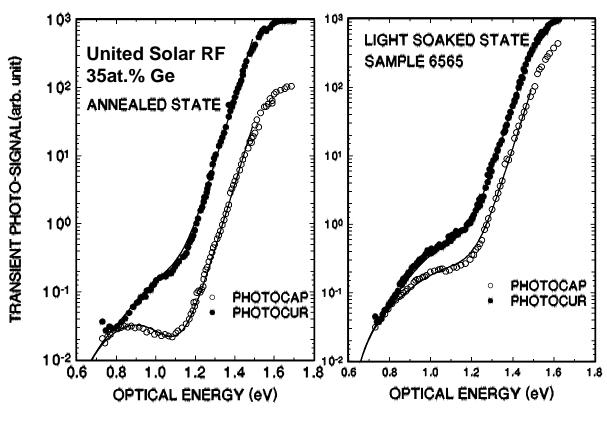


For this United Solar sample the hole collection fraction exceeds 90% under our experimental conditions



## Measuring Fraction of Collected Holes

#### **United Solar RF PECVD 35at.% Ge Sample**



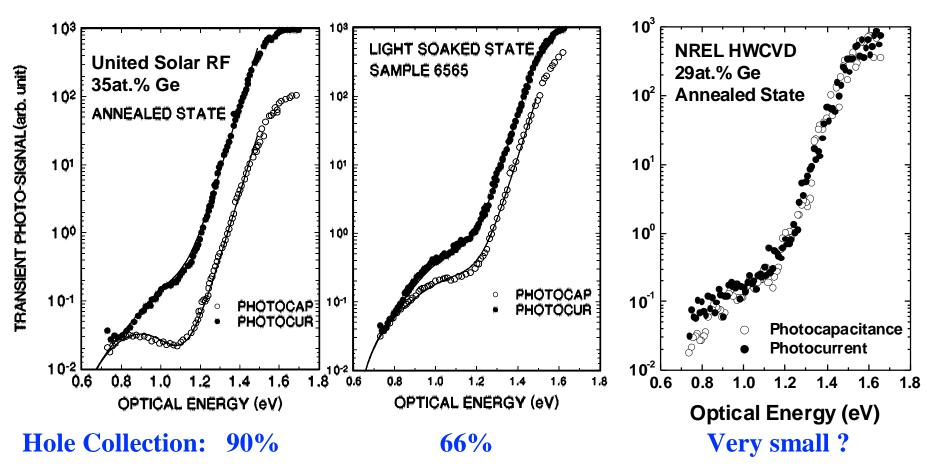
Hole Collection: 90% 66%



## Measuring Fraction of Collected Holes

#### **United Solar RF PECVD 35at.% Ge Sample**

#### NREL HWCVD 29at.% Ge



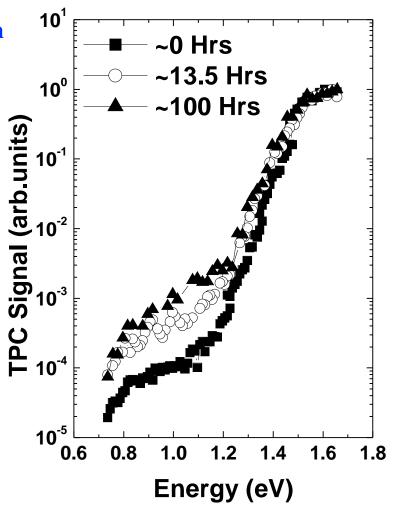


## Light-Induced Degradation

29at.% Ge sample was exposed to 610nm filtered ELH light at 800mW/cm<sup>2</sup>

Increase in deep defect density clearly occurs by a substantial factor

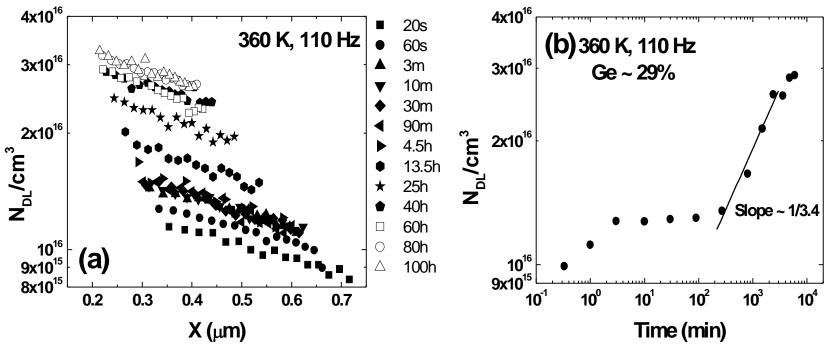
**Unusual creation kinetics are revealed by DLCP studies** 





#### Unusual Defect Creation Kinetics in 29at.% Ge Alloy

Defect density increases by a factor of 1.4 during first 10 minutes and then remains constant for next 3 hours of light exposure before increasing again



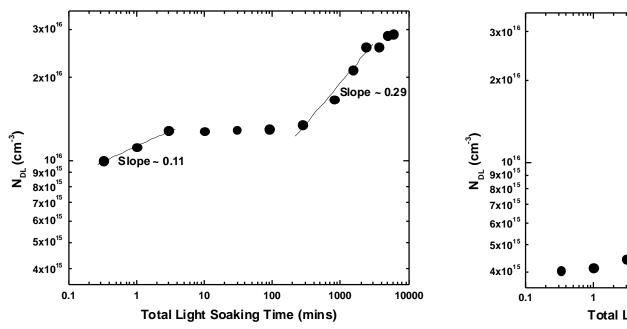


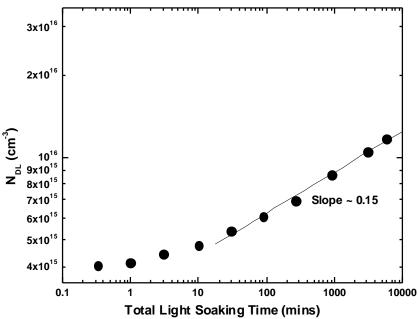


## Comparison with PECVD Alloy Degradation

#### NREL 29at.% Ge HWCVD Alloy

#### **USOC 30at.% Ge PECVD Alloy**





Note: Slightly higher defect levels in HWCVD alloy sample are likely due its slightly smaller energy gap.



## Conclusions

- ➤ We have found superior electronic properties for NREL HWCVD a-Si,Ge:H alloys grown using lower filament temperatures.
- ➤ In particular these samples exhibit sharp band tails and low midgap defect densities, comparable to the best PECVD a-Si,Ge:H samples.
- Electronic properties appear to be very sensitive to oxygen impurity levels, perhaps much more so than PECVD alloy samples.
- ➤ The minority hole collection appears to be less efficient than the best PECVD alloys in apparent contradiction to the above results.
- ➤ Preliminary degradation studies of the 29at.% Ge alloy sample indicate an unusual two-step defect creation kinetics.